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Magnetic phase transitions in Pr_{1-x}Lu_xMn₂Ge₂ compounds

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Abstract

The effects of replacing Pr by Lu on the magnetic behaviour and structures of $Pr_{1-x}Lu_xMn_2Ge_2$ (x = 0.2, x = 0.4) have been investigated using x-ray diffraction, Mössbauer spectroscopy, magnetization and neutron diffraction measurements. The substitution of Lu for Pr leads to a decrease in the lattice constants a, c and the unit cell volume V at room temperature with this contraction of the unit cell resulting in modifications of the $Pr_{1-x}Lu_xMn_2Ge_2$ magnetic structures. Four and five magnetic phase transitions-linked primarily with temperature driven changes in the intralayer Mn-Mn separation distances-have been detected within the temperature range 4.5–550 K for Pr_{0.8}Lu_{0.2}Mn₂Ge₂ and Pr_{0.6}Lu_{0.4}Mn₂Ge₂, respectively, with re-entrant ferromagnetism being detected around $T_{\rm C}^{\rm Pr} \sim 31$ K for ${\rm Pr}_{0.6}{\rm Lu}_{0.4}{\rm Mn}_2{\rm Ge}_2$. It was found that $T_{\rm C}^{\rm inter}$ and $T_{\rm C}^{\rm Pr}$ increase with increasing applied field while $T_{\rm N}^{\rm inter}$ decreases for Pr_{0.6}Lu_{0.4}Mn₂Ge₂, indicating that the canted antiferromagnetic AFmc region contracts with increasing field. The Debye temperatures for $Pr_{1-x}Lu_xMn_2Ge_2$ with x = 0.2 and 0.4 were evaluated as $\theta_D = 320 \pm 40$ K and $\theta_D = 400 \pm 20$ K respectively from the temperature dependence of the average isomer shift. The magnetic structures of both compounds have been determined by means of neutron diffraction measurements over the temperature range 3-300 K with formation of the Fmi magnetic state below $T_{c/c} = 192$ K for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ and the occurrence of re-entrant ferromagnetism below $T_{\rm C}^{\rm Pr} = 31$ K for $\Pr_{0.6} Lu_{0.4} Mn_2 Ge_2$ being confirmed.

(Some figures in this article are in colour only in the electronic version)

1. Introduction

The magnetic properties of the ternary rare earth RMn₂Ge₂ compounds with the tetragonal ThCr₂Si₂-type structure (I4/mmm) have continued to attracted interest in recent years; this is linked in part to their natural multilayer structure which can exhibit either coupled magnetic and crystallographic transitions or valence-related transitions (e.g. [1–5]) and also plays a role in materials which exhibit a giant magnetocaloric effect (GMCE). It is well known that the main driving mechanism for the magnetism of these 1-2-2 type compounds is the strong dependence of Mn–Mn intraplanar and interplanar exchange interactions on d_{Mn-Mn}^{a} , the Mn–Mn separation distance in the *ab*-plane [6, 7]. Recently, it was reported

that several compounds with a distinct layered structure exhibit a giant magnetocaloric effect (GMCE) such as $Gd_5Si_2Ge_2$ [8], $RTi_{1-x}Mo_xGe$ (R = Tb, Er; x = 0, 0.15) [9] and $R_{1-x}M_xMnO_3$ (R = La, Nd, Pr and M = Ca, Sr, Ba, etc) [10], where interlayer distances play a vital role in determining their magnetic phase transition order/type and in governing the MCE performance. The strong dependence of the interlayer exchange interaction in RMn₂Ge₂ on the distance between magnetic atoms in layered compounds [2, 11, 12] is expected to offer scope for selection of the magnetic state and therefore the MCE—by controlling the interlayer and intralayer distances between magnetic atoms using chemical (i.e. brought about by compositional changes) or external pressure [13]. Indeed, recently it has been confirmed that

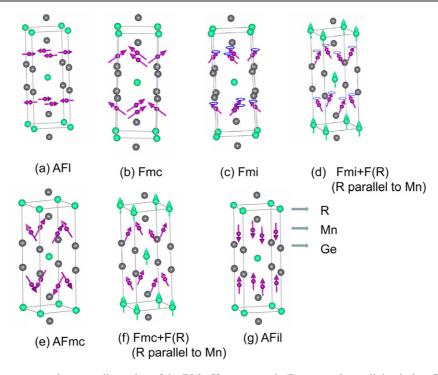


Figure 1. The magnetic structures relevant to discussion of the RMn_2X_2 compounds (R atoms—larger light circles; Ge atoms—larger dark circles; Mn atoms—small circles). The different structure types are defined in [7].

the Sm–Mn coupling plays a crucial role in determining the magnitude of magnetocaloric properties in $SmMn_2Ge_2$ both at ambient and under applied pressures [14].

Here, we investigate the effects on the magnetic structure of substituting Lu for Pr in $Pr_{1-x}Lu_xMn_2Ge_2$ (x = 0.2, 0.4) for the first time using magnetic, Mössbauer spectroscopy and neutron diffraction measurements. In addition to exploring the effects of the nonmagnetic Lu atoms on the rare earth Pr sublattice, it is expected that substitution of Pr (atomic radius 1.82 Å) with the smaller Lu (atomic radius 1.75 Å) would modify the magnetic structure of $Pr_{1-x}Lu_xMn_2Ge_2$. This follows as $d^a_{\rm Mn-Mn} \sim 2.915$ Å in PrMn₂Ge₂ [15] while $d_{\rm Mn-Mn}^a \sim 2.785$ Å in LuMn₂Ge₂ [7]. These values of $d_{\rm Mn-Mn}^a$ are respectively greater than $d_{\rm crit1} \sim 2.87$ Å and less than $d_{\rm crit2} \sim 2.84$ Å, the first and second critical intralayer Mn–Mn distances which govern the magnetic behaviour in RMn₂Ge₂ compounds [6, 7]. According to Welter *et al* [15], the d_{Mn-Mn}^a not only affects the intralayer Mn-Mn coupling but also the interlayer exchange interaction and three general categories can be delineated [6, 13].

- (i) The interlayer exchange coupling being ferromagnetic and the intralayer coupling antiferromagnetic for $d_{Mn-Mn}^a > d_{crit1} = 2.87$ Å leading to the canted Fmc-type ferromagnetic structure.
- (ii) Both the interlayer and the intralayer coupling being antiferromagnetic (AFmc-type magnetic structure) for $d_{\text{crit2}} = 2.84 \text{ Å} < d_{\text{Mn-Mn}}^a < d_{\text{crit1}} = 2.87 \text{ Å}.$
- (iii) No intralayer in-plane spin component and the interlayer coupling being antiferromagnetic, leading to the AFil-type magnetic structure d^a_{Mn-Mn} in the range $d^a_{Mn-Mn} < d_{crit2} = 2.84$ Å [6, 13].

In order to aid discussion of the magnetic behaviours observed in this study, several of the relevant Mn sublattice magnetic structures are depicted in figure 1 with the notation used to describe the magnetic structure types and critical transition temperatures defined in [7]. Magnetic transition from paramagnetic (P) to intralayer antiferromagnetic ordering within the (001) Mn layers (AFl) at $T_{\rm N}^{\rm intra} = 415$ K has been reported in PrMn₂Ge₂ (lattice parameter a = 4.123 Å at room temperature) [15] and below $T_{\rm C}^{\rm inter}$ = 334 K a canted spin structure (Fmc) with the c-axis interlayer spin components aligning parallel and antiferromagnetic ordering within the (001) Mn layers is formed. With further decrease in temperature to below $T_{c/c} = 280$ K the structure transforms from Fmc to a conical configuration Fmi type (ferromagnetic mixed incommensurate structures characterised by the ferromagnetic interplane coupling of the in-plane ferromagnetic components and by the incommensurate arrangement of the antiferromagnetic in-plane components) and below about 100 K, the Pr sublattice in PrMn₂Ge₂ orders magnetically along the c-axis with the Pr moments coupled parallel to the Mn moments (Fmi + F(Pr)). With LuMn₂Ge₂ (lattice parameter a = 3.939 Å at room temperature) found to exhibit the antiferromagnetic interlayer coupling structure AFil below $T_{\rm N} = 508$ K [7], it is clear that a variety of interesting magnetic phenomena can be expected on substituting Lu for Pr in $Pr_{1-x}Lu_xMn_2Ge_2$. Here we investigate $Pr_{1-x}Lu_xMn_2Ge_2$ compounds with x = 0.2 and 0.4, using magnetic, Mössbauer and neutron diffraction measurements.

2. Experimental details

 $Pr_{1-x}Lu_xMn_2Ge_2$ (⁵⁷Fe) compounds with x = 0.2 and 0.4 were prepared by conventional argon arc melting. The starting

Table 1. Lattice parameters and structural information for the $Pr_{1-x}Lu_xMn_2Ge_2$ compounds at room temperature (the errors for the lattice parameters *a* and *c*, and the interatomic spacing distances were derived to be ± 0.001 Å from the data fits; z_{Ge} : ± 0.001). $\Delta V/V_0 = (V - V_0)/V_0$ where *V* and V_0 are the volume of the unit cell volume for $Pr_{1-x}Lu_xMn_2Ge_2$ and $PrMn_2Ge_2$, respectively.

x	d _{Mn-Mn} (Å)	d _{Mn-Ge} (Å)	$d_{ m Mn-Pr}$ (Å)	d _{Pr-Ge} (Å)	d _{Pr-Pr} (Å)	d _{Ge-Ge} (Å)	a (Å)	<i>c</i> (Å)	$V(\text{\AA}^3)$	Z.	$\Delta V/V_0$ (%)	Reference
0	2.915	2.510	3.422	3.192	4.123	2.600	4.123	10.925	185.7	0.381		[15]
0.2	2.893	2.511	3.407	3.158	4.091	2.537	4.091	10.898	182.4	0.384	1.78	
0.4	2.872	2.497	3.393	3.138	4.061	2.497	4.061	10.873	179.3	0.385	3.45	
1.0	2.785				3.939		3.939	10.809	167.7		9.69	[7]

materials contained $\sim 3\%$ excess Mn to compensate for the Mn loss due to evaporation during melting and the ingot was melted five times for improved homogeneity. The samples were doped with $\sim 0.5 \text{ wt}\%^{57}$ Fe for the Mössbauer studies and the samples were characterized by x-ray diffraction (Cu K α radiation; $\lambda = 1.5418$ Å). The temperature dependence of the magnetization, M(T), was measured in a magnetic field of $B_{\text{appl}} = 0.005 \text{ T}$ in a superconducting quantum interference device (SQUID) from 5 to 350 K. The differential scanning calorimetry (DSC) measurements have been measured to check for possible phase transitions in the higher temperature range from 300 to 550 K. The transition temperatures $T_{\rm C}^{\rm Pr}$ and $T_{\rm N}^{\rm inter}$ for compounds were defined as the temperatures at which dM/dT exhibits an extreme value [16] while the magnetic transition temperatures $T_{\rm C}^{\rm inter}$ were derived from the thermal magnetic curves by extrapolating M^2-T plots to $M^2 = 0$ [17]. The ⁵⁷Fe Mössbauer spectra were obtained between 4.2 and 298 K using a standard constant-acceleration spectrometer and a ⁵⁷CoRh source. The spectrometer was calibrated at room temperature with an α -iron foil. Neutron diffraction experiments were carried out on the powder samples using the MRPD diffractometer ($\lambda = 1.6660$ Å) of Reactor HIFAR (shut down in January 2007 after 50 years of safe operation), Lucas Heights, and the high intensity diffractometer Wombat $(\lambda = 2.410 \text{ Å})$ of Australia's Replacement Research Reactor OPAL, Lucas Heights at temperatures over the range 3-300 K.

3. Results and discussion

X-ray diffraction patterns indicate that the compounds have a single phase with the tetragonal ThCr₂Si₂-type structure (space group I4/mmm, space group number 139). The patterns have been analysed using the FULLPROF software and the lattice parameters have been obtained (listed in table 1). The replacement of Pr with Lu leads to the expected decrease in lattice parameters due to smaller atomic size of Lu than Pr. The bond lengths between different sites have also been calculated with the BLOKJE program [18] using the structural and positional parameters and the 12-coordinate metallic radii of 1.82 Å, 1.75 Å, 1.35 Å and 1.37 Å for Pr, Lu, Mn and Ge, respectively.

It can be seen from table 1 that, with increasing Lu content, the intralayer Mn–Mn spacing d_{Mn-Mn}^a at room temperature for Pr_{1-x}Lu_xMn₂Ge₂ decreases but is still larger than the first critical value of 2.87 Å for the samples with x = 0.2 and 0.4. If we assume that the contraction of the unit cell volume due to Lu substitution (chemical

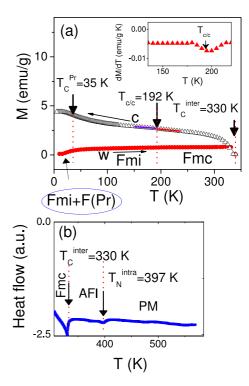


Figure 2. The temperature dependence of magnetization for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ over the range 3–350 K as measured in a magnetic field of 0.005 T on warming (W) after first zero field cooling (ZFC). Measurements were then obtained in a magnetic field of 0.005 T on cooling (C). The inset to (a) shows the dM/dT versus *T* curve indicating how $T_{c/c}$ was determined. (b) DSC measurements for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ over the temperature range 300–550 K.

pressure) is equivalent to the influence of external pressure, the corresponding pressures can be derived to be 15.2 and 30.8 kbar for the x = 0.2 and 0.4 samples using the Murnaghan equation $p = (B_0/B'_0)[(V/V_0)^{-B'_0} - 1]$ where B_0 is the isothermal bulk modulus, B'_0 its pressure derivative and V_0 and V are the volume at ambient pressure and pressure p respectively. In this calculation we used the B_0 (=819 kbar) and B'_0 (=4.0) values of LaMn₂Si₂ [19] to approximate the values for PrMn₂Ge₂.

3.1. Magnetic properties

 $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$. Figure 2(a) shows the temperature dependence of the magnetization of $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ as measured in a field of 0.005 T over the temperature range of 5–350 K after cooling in zero field (the warming and cooling curves are marked by arrows). It can be seen clearly that below

350 K two magnetic phase transitions are readily detected at \sim 330 and \sim 35 K with indications of a third possible phase transition around \sim 192 K as evidenced by a slight change in the slope of the *M* versus *T* curve (as marked by an arrow; this transition around 192 K is more evident in the d*M*/d*T* curve shown as an insert to figure 2(a)). DSC measurements (shown as figure 2(b)) have been used to check for possible phase transition in the temperature region up to 550 K. Two transitions at around \sim 330 and \sim 397 K have been detected with the transition at around \sim 330 K agreeing well with the observations from the region of overlap with the magnetization measurements.

In order to determine the type and order of these transitions, we have measured the field dependence of magnetization (figure 3(a)) at selected temperatures and, based on these magnetic data, the Arrot-plot (M^2 versus H/M curves [20]) are drawn in figure 3(b). It is clear that, with a positive intercept on the M^2 axis extrapolated from high field, the compound exhibits ferromagnetic character at 300, 100 and 10 K while $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ behaves like an antiferromagnet or paramagnet at 340 K where a negative intercept on the M^2 axis is obtained.

In comparison with the behaviour of the limiting $PrMn_2Ge_2$ compound which exhibits the antiferromagnetic AFl structure (figure 1(a)) between 415 and 334 K and the ferromagnetic Fmc structure (figure 1(b)) between 334 and 280 K [15], it appears reasonable for us to ascribe the transition at ~330 K in $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ to a transition from the AFl structure to the Fmc structure while the transition at 35 K can be assumed to be the additional ferromagnetic contribution stemming from ordering of the Pr sublattice at T_C^{Pr} . Likewise, the transition at ~192 K should be assigned to be $T_{c/c}$ from the high temperature canted Fmc to the lower temperature conical spin structure, Fmi (figure 1(c)) while the transition at ~397 K corresponds to the transition from paramagnetism to AFl antiferromagnetism. These conclusions are considered further in discussion of the neutron diffraction results below.

As stated above, as a general guide, if $d_{Mn-Mn}^a > 2.87$ Å, the intralayer in-plane coupling is antiferromagnetic and the interlayer coupling is ferromagnetic [6, 7]. The magnetic behaviour of Pr_{0.8}Lu_{0.2}Mn₂Ge₂ can therefore be understood generally in these terms as d_{Mn-Mn}^a at room temperature is 2.893 Å which is larger than the first critical value $d_{crit1} \sim 2.87$ Å discussed above. Moreover, the values of spontaneous magnetization M_s (and corresponding magnetic moment values per formula unit) have been calculated from the Arrott-plot to be: 58.1 emu g⁻¹ (4.2 $\mu_B/f.u.$), 41.5 emu g⁻¹ (3.0 $\mu_B/f.u.$) and 26.3 emu g⁻¹ (1.9 $\mu_B/f.u.$) at 10 K, 100 K and 300 K, respectively.

 $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$. Compared with $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$, the value $d^a_{Mn-Mn} = 2.872$ Å at room temperature for $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ is similar to $d_{crit1} \sim 2.87$ Å. With decreasing temperature, the unit cell will contract and the magnetic interaction between the Mn layers is therefore expected to change from ferromagnetic to antiferromagnetic. Previous investigations [6, 7] of RMn_2Ge_2 indicate that when 2.84 Å < d^a_{Mn-Mn} < 2.87 Å, the intralayer in-plane coupling becomes AF and the interlayer coupling is also

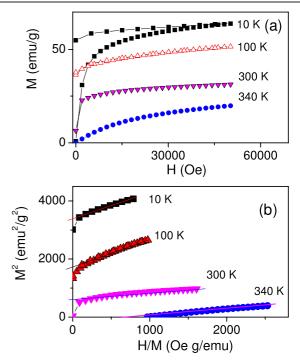


Figure 3. (a) The variation of magnetization with applied magnetic field ($H_{appl} = 0-50 \text{ kOe}$) for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ at the temperatures indicated. (b) The Arrot-plots (M^2 versus M/H) derived from the magnetization data of (a).

AF. In fact, similar to the case of SmMn₂Ge₂ [11], reentrant ferromagnetism of the Mn sublattice [12, 13] has been observed in Pr_{0.6}Lu_{0.4}Mn₂Ge₂ below 31 K as shown in the magnetization data (figure 4(a); T_c^{Pr} and T_N^{inter} were determined from the dM/dT versus T curves shown as the inserts to figure 4(a)). This ferromagnetic ordering of the Mn sublattice is driven by the ferromagnetic ordering of the Pr sublattice which takes place at $T_{\rm C}^{\rm Pr} = 31$ K. Likewise [21], as indicated by the magnetization data of figure 4(a) and the DSC data of figure 4(b), the transition temperatures have been determined to be ~321 K, ~172 K and ~31 K for $T_{\rm C}^{\rm inter}$, $T_{\rm N}^{\rm inter}$ and $T_{\rm C}^{\rm Pr}$, respectively. Figure 5(a) shows a set of magnetization versus temperature curves for Pr_{0.6}Lu_{0.4}Mn₂Ge₂ at various fields with the field dependence of the critical temperatures $T_{\rm C}^{\rm inter}$, $T_{\rm N}^{\rm inter}$ and $T_{\rm C}^{\rm Pr}$ shown in figure 5(b). As indicated in figure 5(b), $T_{\rm N}^{\rm inter}$ decreases and $T_{\rm C}^{\rm Pr}$ increases as the magnetic field increases, resulting in a narrowing of the canted AFmc antiferromagnetic The field dependence of these magnetic phase range. transition temperatures is derived to be: $dT_{\rm C}^{\rm inter}/\mu_0 dH$ = 3.9 K T⁻¹; $dT_{\rm N}^{\rm inter}/\mu_0 dH = -8.6$ K T⁻¹, and $dT_{\rm C}^{\rm Pr}/\mu_0 dH =$ 3.5 K T^{-1} with fitting results shown by the lines in figure 5(b). Similar behaviours have been found in other systems with re-entrant ferromagnetism including: SmMn₂Ge₂ [12]; NdMn_{1.575}Fe_{0.425}Ge₂ (where $dT_{N}^{inter}/\mu_{0}dH = -13 \text{ K T}^{-1}$ and $dT_{C}^{R}/\mu_{0}dH = 8 \text{ K T}^{-1}$ [13]; PrMn_{0.6}Fe_{0.4}Ge₂ [21], and $Nd_{0.35}La_{0.65}Mn_2Si_{2}$ [7, 22]. On the other hand, it was reported that the antiferromagnetic range in SmMn₂Ge₂ [14, 23] becomes wider when hydrostatic pressure is applied $(dT_C^R/\mu_0 dH)$ $= -9.1 \text{ K kbar}^{-1}$ and $dT_{\text{N}}^{\text{inter}}/\mu_0 dH = 22.3 \text{ K kbar}^{-1}$ [14]. This indicates that the influence of an applied magnetic field on

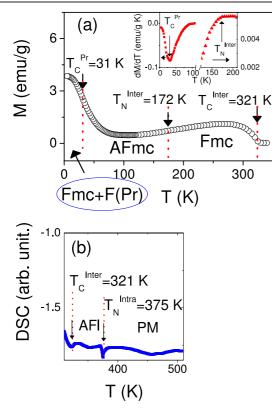


Figure 4. (a) The temperature dependence of magnetization for $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ as measured in a magnetic field of 0.005 T on warming after zero field cooling (ZFC). The insets to (a) show the dM/dT versus T curves indicating how $T_{\rm N}^{\rm Pr}$ and $T_{\rm N}^{\rm inter}$ were determined. (b) DSC measurements over the temperature range 300–550 K.

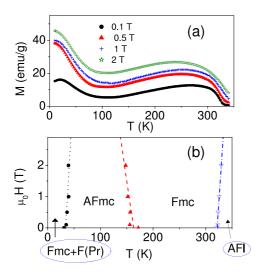


Figure 5. (a) The temperature dependence of magnetization for $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ at the magnetic field values indicated, and (b) a partial magnetic phase diagram for $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$. The dashed lines represent fits to the data as described in the text.

the magnetic phase transitions is opposite to that from external pressure [24, 25]; this behaviour can be understood in terms of the spontaneous magnetovolume effect observed in these systems [7, 21]. Moreover, the transition at $T_{\rm N}^{\rm intra} \sim 375$ K from paramagnetism (PM) to *ab*-plane antiferromagnetism,

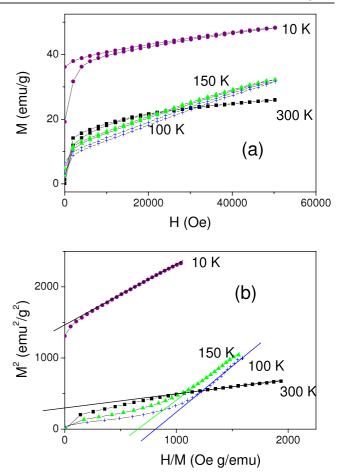


Figure 6. (a) The variation of magnetization with applied magnetic field ($H_{appl} = 0-50$ kOe) for $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ at the temperatures indicated and (b) the Arrot-plots (M^2 versus M/H) derived from the magnetization data of (a).

AFl, has been detected from the DSC measurement for high temperature range (300-550 K) as shown in figure 4(b).

The field dependences of the magnetization of Pr_{0.6}Lu_{0.4} Mn_2Ge_2 at selected temperature are shown in figure 6(a) with the corresponding Arrot-plots drawn in figure 6(b). These temperatures-10, 100, 150, and 300 K-were selected as being representative of the different magnetic regions indicated by the magnetization versus temperature curve of figure 4(a). The M^2 versus H/M plots indicate that $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ exhibits ferromagnetic behaviour at 10 and 300 K (positive intercepts on the M^2 axis) and antiferromagnetic behaviour at 100 and 150 K (negative intercepts on the M^2 axis). These findings have been confirmed by our neutron diffraction investigations as discussed in section 3.3 below. Further, in the ferromagnetic regions the M_s and magnetic moment values per formula unit values have been derived to be 38.3 emu g^{-1} (2.8 $\mu_{\rm B}/{\rm f.u.}$) and 17.2 emu g⁻¹ (1.3 $\mu_{\rm B}/{\rm f.u.}$) at 10 K and 300 K, respectively.

Compared with PrMn₂Ge₂ (5.28 $\mu_B/f.u.$ at 1.5 K), the decrease in M_s at 10 K for Pr_{0.6}Lu_{0.4}Mn₂Ge₂ (4.2 $\mu_B/f.u.$) and Pr_{0.8}Lu_{0.2}Mn₂Ge₂ (2.8 $\mu_B/f.u.$) can be linked with dilution of the Pr magnetic moment with nonmagnetic Lu. Assuming that the Mn magnetic moments below T_C^{Pr} are same in these

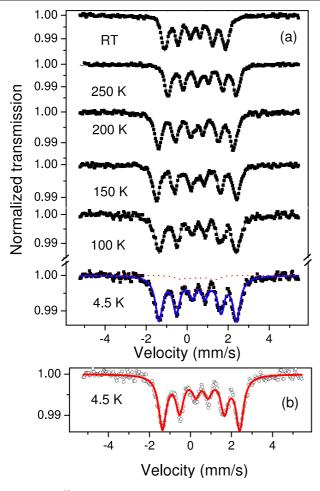
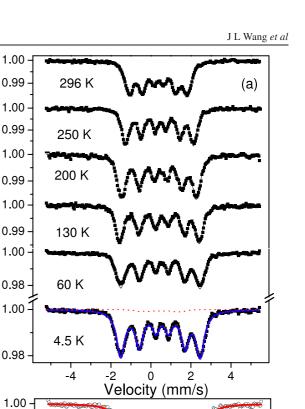


Figure 7. (a) ⁵⁷Fe Mössbauer spectra of $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ over the temperature range 4.5–300 K. The spectra were fitted using two sextets. The sub-spectral component (sextet II; dashed line in the 4.5 K spectrum) indicates the fraction (\sim 7(2)%) of ⁵⁷Fe atoms located in the Ge 4e site as discussed in the text. The remainder of the ⁵⁷Fe atoms (sextet I; \sim 93(2)%) are located in the Mn 4d site. (b) ⁵⁷Fe Mössbauer spectrum of $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ at 4.5 K fitted using one sextet.

three compounds, M_s is expected to decrease with increase in Lu content as, below the rare earth ordering temperature T_c^R , the Mn moments order ferromagnetically for light rare earths in RMn₂X₂ compounds (but antiferromagnetically for the heavy rare earths). However, as is well understood, the Mn moment itself is very sensitive to its local environment including the spacing, number and size of neighbouring atoms. As the substitution of Lu for Pr clearly shortens the spacing of the Mn–Mn and Mn–Pr atoms (table 1), a decrease in the value of the Mn moment due to the decrease of d_{Mn-Mn}^a may also play a role in the overall decrease in M_s with increasing Lu content. Similar effects have been observed in the GdMn₂Ge_{2-x}Si_x system [22] where a decrease in the Mn moment is detected as x is increased from 0 to 2.

The magnetization behaviour exhibited by $Pr_{0.6}Lu_{0.4}Mn_2$ Ge₂ in figure 4(a) is similar to those of the re-entrant ferromagnetic systems Nd_{0.35}La_{0.65}Mn₂Si₂ [22], Pr_{1-x}Y_xMn₂Ge₂ [16] and SmMn₂Ge₂ [23]. This allows comparison of these related 1-2-2 systems in order to assess the influence of geometrical



Normalized transmission

Figure 8. (a) ⁵⁷ Fe Mossbauer spectra of $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ over the temperature range 4.5–300 K. The spectra were fitted using two sextets. The sub-spectral component (sextet II; dashed line in 4.2 K spectrum) indicates the fraction (~4(2)%) of ⁵⁷Fe atoms located in the Ge 4e site as discussed in the text. The remainder of the ⁵⁷Fe atoms (sextet I; ~96(2)%) are located in the Mn 4d site. (b) ⁵⁷Fe Mössbauer spectrum of $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ at 4.5 K fitted using one sextet.

factors in governing their behaviour. For example, the room temperature lattice parameter of $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$, a = 4.0613 Å, is slightly smaller than that of $Nd_{0.35}La_{0.65}Mn_2Si_2$, $a \sim 4.070$ Å [22], $Pr_{0.5}Y_{0.5}Mn_2Ge_2$, a = 4.083 Å and $Pr_{0.4}Y_{0.6}Mn_2Ge_2$, a = 4.069 Å [16], but larger than that of SmMn_2Ge_2, $a \sim 4.045$ Å [23]. Moreover, compared with these $R_{1-x}R'_xMn_2Ge_2$ systems, the critical lattice parameters in the $RMn_{2-x}Fe_xGe_2$ system for which re-entrant ferromagnetism is also found to occur, are relatively larger (for example, a = 4.0811 Å in $NdFe_{0.425}Mn_{1.575}Ge_2$ [13] and a = 4.088 Å in $PrMn_{1.4}Fe_{0.6}Ge_2$ [21]). This common reentrant magnetic behaviour for disparate values of the lattice parameter a indicates that geometric factors alone do not govern the magnetic behaviour and that electronic effects also play a role [21].

3.2. Mössbauer spectroscopy

Typical Mössbauer spectra for the $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ and $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ samples at various temperatures between

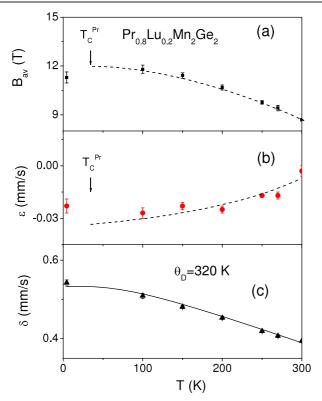


Figure 9. Temperature dependences of the Mössbauer hyperfine parameters (sextet I) of $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$: (a) average magnetic hyperfine field B_{hf} ; (b) quadrupole shift ε (c) isomer shift δ . The line in (c) represents the fit of the isomer shift data to equation (2) resulting in the Debye temperature $\theta_D = 320 \pm 20$ K. The dashed lines in (a) and (b) represent the trends of the data.

4.5 K and 300 K are shown in figures 7(a) and 8(a) respectively. The ⁵⁷Fe atoms are expected to predominantly enter the Mn crystallographic 4d site and ideally the spectra would be fitted using a single sextet to represent ⁵⁷Fe atoms in the 4d site. In practice, and in agreement with Mössbauer studies of related 1-2-2 compounds, two sextets are generally required to provide the best spectral fits (e.g. [26, 27]). In the present investigation, the fits of the 4.5 K spectra for Pr_{0.8}Lu_{0.2}Mn₂Ge₂ and $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ using one sextet (figures 7(b) and 8(b) respectively) are shown for comparison with the fits obtained using two sextets (figures 7(a) and 8(a) respectively). These comparisons demonstrate that significant misfits remain when only one sextet is used and highlight the need to include the second sextet for proper representation of the spectra. These two sextets represent ⁵⁷Fe atoms that predominantly enter the Mn 4d site (sextet I) and those which enter the Ge 4e site (sextet II). The optimal fits to the present spectra reveal that the large majority of ⁵⁷Fe atoms reside on the 4d site (93(2) % and 96(2) % for Pr_{0.8}Lu_{0.2}Mn₂Ge₂ and Pr_{0.6}Lu_{0.4}Mn₂Ge₂, respectively) with the remainder of the 57 Fe atoms (7(2) % and 4(2) % respectively) residing on the 4e site. This behaviour is also consistent with the fact that sextet I of the Mn 4d site exhibits a larger magnetic hyperfine field than sextet II of the Ge 4e site (e.g. for Pr_{0.8}Lu_{0.2}Mn₂Ge₂ at 4.5 K the hyperfine fields are: $B_{\rm hf}(4d) = 11.7$ T; and $B_{\rm hf}(4e) = 5.9$ T) (cf e.g [27]). Examples of the sub-spectral components for

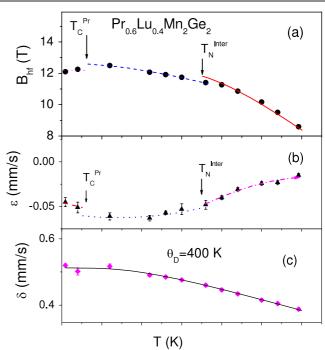


Figure 10. Temperature dependences of the Mössbauer hyperfine parameters (sextet I) of $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$: (a) average magnetic hyperfine field $B_{\rm hf}$; (b) quadrupole shift ε (c) isomer shift δ . The line in (c) represents the fit of the isomer shift data to equation (2) resulting in the Debye temperature $\theta_{\rm D} = 400 \pm 20$ K. The dashed and full lines in (a) and the dashed line in (b) represent the trends of the data.

 $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ and $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ are shown in the 4.5 K spectra of figures 7(a) and 8(a) respectively.

The temperature dependences of the Mössbauer hyperfine parameters for Pr_{0.8}Lu_{0.2}Mn₂Ge₂ and Pr_{0.6}Lu_{0.4}Mn₂Ge₂—as represented by the behaviour of the 57Fe atoms located on the Mn 4d site (sextet I)-are shown in figures 9 and 10 respectively. As is well accepted, the observed hyperfine fields result from contributions to the exchange interactions present in the Pr and Mn sublattices and correspondingly reflect the magnetic order in the Pr and Mn sublattices [26-28]. The variation in the hyperfine field values around the transition temperatures determined from the magnetic measurements (figures 2(a) and 4(a)) is clearly linked with modifications of the magnetic structures. It can be seen from figure 9 that, surprisingly, the magnetic hyperfine field for Pr_{0.8}Lu_{0.2}Mn₂Ge₂ at 4.5 K (below the ordering temperature for the Pr sublattice of $T_{\rm C}^{\rm Pr} \sim 35$ K; figure 2(a)) is less than the magnetic hyperfine field at 100 K even though the Pr and Mn moments couple ferromagnetically below $T_{\rm C}^{\rm Pr}$ as observed in other similar systems such as NdMn₂Ge₂ [26]. This implies that the transferred hyperfine field from the rare earth sublattice is opposite to that contributed from the Mn sublattice [21, 26]. For $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$, the transition temperatures T_C^{Pr} and $T_{\rm N}^{\rm inter}$ are clearly marked by changes in the trends of ⁵⁷Fe magnetic hyperfine field and the electric quadrupole interaction parameters as shown in figure 10. Similar behaviour has been reported in other systems including $PrMn_{2-x}Fe_xGe_2$ [21], $La_{1-x}Y_xMn_2Si_2$ [27] and RMn_2X_2 with X = Ge and Si [26].

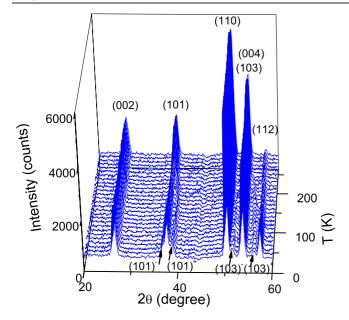


Figure 11. Representative neutron diffraction patterns for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ over the temperature range 5–265 K at 10 K intervals as obtained on the high intensity diffractometer Wombat, OPAL ($\lambda = 2.410$ Å).

The Debye temperatures for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ and $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$, have been determined from the temperature dependence of the isomer shift, IS(T) [29, 30]:

$$IS(T) = IS_0(T) + IS_{SODS}(T).$$
 (1)

 $IS_0(T)$ represents the *T* dependence of the charge density at the probe nucleus which is generally weakly temperature dependent. The second term, IS_{SODS} , is the so-called secondorder Doppler shift, which, using the Debye model for the phonon spectrum, can be described by,

$$IS_{SODS}(T) = -\frac{3kT}{2mc} \left[\frac{3\theta_{\rm D}}{8T} + 3\left(\frac{T}{\theta_{\rm D}}\right)^3 \int_0^\tau \frac{x^3}{e^x - 1} \,\mathrm{d}x \right],\tag{2}$$

where m is the mass of the ⁵⁷Fe nucleus, k Boltzmann's constant, c the velocity of light, and $\tau = \theta_{\rm D}/T$ the reduced temperature [29, 30]. Debye temperature values of θ_D = 320 ± 20 K and $\theta_{\rm D} = 400 \pm 20$ K have been derived for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ and $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ respectively on fitting the mean isomer shift data of figures 9(c) and 10(c)to equation (2). These Debye temperatures are similar to $\theta_{\rm D}$ values reported for RMn₂X₂ systems. For example, the value of $\theta_D = 338$ K has been obtained for YMn₂Ge₂ on fitting the thermal expansion [31], while the value of $\theta_{\rm D} = 415$ K for Gd_{1-x}La_xMn₂Ge₂ and YCo₂Ge₂ has been obtained on fitting the temperature dependence of lattice parameters [32]. In the present case, the tendency for increase in the Debye temperature with increasing Lu content in Pr_{0.8}Lu_{0.2}Mn₂Ge₂ and Pr_{0.6}Lu_{0.4}Mn₂Ge₂, can be understood in the term of the difference of Debye temperature between Pr ($\theta_D = 85$ K) and Lu ($\theta_{\rm D} = 185 \text{ K}$) [33].

Table 2. Structural and magnetic parameters derived from Rietveld refinements of the neutron diffraction patterns for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ compound (cf figure 12). The errors were derived from the data fits (the errors are shown for the 3 K data).

T (K)	3	60	220	290
a (Å)	4.0726(3)	4.0734(3)	4.0810(4)	4.0852(5)
<i>c</i> (Å)	10.8589(14)	10.8589(14)	10.8748(18)	10.8869(20)
ZGe	0.3823(3)	0.3828(2)	0.381(3)	0.3814(4)
μ_{ab} ($\mu_{\rm B}$)	2.71(7)	2.68(6)	1.09(3)	1.04(3)
$\mu_Z (\mu_B)$	2.10(7)	2.07(6)	1.36(1)	1.36(1)
$\mu_{\text{total}} \left(\mu_{\text{B}} \right)$	3.43(7)	3.38(6)	1.74 (2)	1.71(2)
q_z	0.215(2)	0.196(2)		
$\mu_{\rm Pr}$ ($\mu_{\rm B}$)	3.13 (9)	_	_	
R_{wp}	6.41	6.35	7.26	7.14
$R_{\rm exp}$	1.52	1.51	1.53	1.54

Table 3. Structural and magnetic parameters derived from Rietveld refinement of neutron diffraction patterns at selected temperatures for $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ (cf figure 14). The errors were derived from the data fits (the errors are shown for the 3 K data).

T (K)	7	100	300
a (Å)	4.0514(13)	4.0468(15)	4.0673(1)
c (Å)	10.8499(39)	10.8563(43)	10.8838(26)
ZGe	0.3832(7)	0.383(1)	0.382(1)
$\mu_{ab} \left(\mu_{\mathrm{B}} \right)$	2.03 (5)	1.65(15)	1.54(11)
$\mu_Z (\mu_{\rm B})$	2.12(7)	1.26(26)	1.31(24)
$\mu_{\text{total}} \left(\mu_{\text{B}} \right)$	2.94(9)	2.08(20)	2.02 (19)
$\mu_{\mathrm{Pr}}\left(\mu_{\mathrm{B}}\right)$	1.98(9)	_	_
R_{wp}	1.98	2.09	2.79
R _{exp}	3.70	3.55	4.62

3.3. Neutron diffraction

Based on the magnetic regions identified in the magnetization measurements for Pr_{0.8}Lu_{0.2}Mn₂Ge₂ and Pr_{0.6}Lu_{0.4}Mn₂Ge₂ (figures 2(a) and 4(a) respectively), we have measured neutron diffraction patterns for both samples at selected temperatures where different magnetic structures are expected. Figure 11 shows a representative series of neutron diffraction patterns taken for the Pr_{0.8}Lu_{0.2}Mn₂Ge₂ compound over the temperature range 5-265 K with 10 K temperature steps. As outlined previously (e.g. [15]), the special position of the Mn atoms on the 4d sites means that the scattering due to ferromagnetic ordering of the Mn atoms contributes only to reflections obeying the (*hkl*) conditions h + k = 2n and l = 2n while the intensity change in the (*hkl*) reflections with h + k = 2n + 1 corresponds to the magnetic contribution from the Mn antiferromagnetic ordering within the (001) planes. Moreover, reflections of the type h + k + l = 2n + 1are based on a collinear antiferromagnetic structure between the adjacent Mn planes. Rietveld refinements have been carried out using the FULLPROF program package which allows simultaneous refinement of the structural and magnetic parameters. Typical examples of the Rietveld refinements to neutron diffraction patterns for Pr_{0.8}Lu_{0.2}Mn₂Ge₂ and Pr_{0.6}Lu_{0.4}Mn₂Ge₂ are shown in figures 12 and 14 respectively with the refinement results listed in tables 2 and 3.

 $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$. Figures 12(a), (b), (c) and (d) show neutron diffraction patterns for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ at 3 K, 60 K, 220 K and 290 K respectively with the temperature variations of the

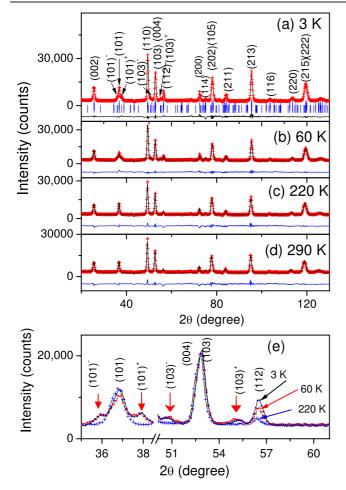


Figure 12. (a), (b), (c), (d) Plots of Rietveld refinements to the neutron diffraction patterns for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ at 3 K, 60 K, 220 K and 290 K, respectively ($\lambda = 2.410$ Å; Wombat, OPAL). The lines show the calculated profiles through the experimental data and the residuals. The markers for the crystallographic (top) and magnetic (bottom) structures are indicated for the 3 K pattern in (a). Figure (e) shows expanded 2θ ranges for the 3, 60 and 220 K patterns.

(112), (101) and (101)⁺ intensities shown in figure 13(a). Consistent with the magnetization results discussed above, the magnetic structure of $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ at 220 and 290 K has been determined to be a canted spin structure (Fmc; figure 1(b)) with the *c*-axis interlayer spin components aligned parallel and the intralayer components exhibiting antiferromagnetic coupling. As is well known [15], the Fmc magnetic structure in RMn₂Ge₂ systems is characterized by magnetic scattering (in addition to nuclear scattering) at the (101) and (112) peaks. The magnetic scattering at the (101) peak corresponds to the contribution from antiferromagnetic ordering within the (00*l*) Mn planes while the (112) reflection corresponds to the ferromagnetic component of the Mn moments along the *c*-axis.

As can be seen in figures 12(b) and (c), the neutron diffraction patterns of $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ at 60 and 3 K feature the satellites peaks (101)⁺ and (101)⁻ as well as (103)⁺ and (103)⁻. The intensities of these satellites are shown for comparison with the 220 K pattern in figure 12(e) (i.e. above the magnetic transition temperature $T_{c/c} = 192$ K of figure 2(a)). This satellite behaviour indicates that at 60 K,

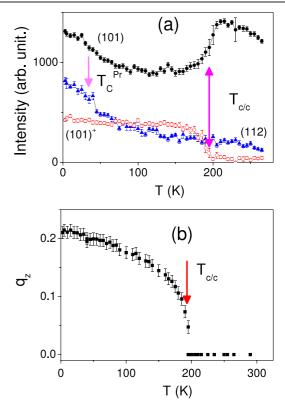


Figure 13. Temperature dependences of: (a) the (101), (112) and (101)⁺ peak intensities and (b) the q_z component of the wavevector for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ as discussed in the text.

Pr_{0.8}Lu_{0.2}Mn₂Ge₂ exhibits a ferromagnetic mixed incommensurate structure (Fmi; see figure 1(c)) of wavevector $(0, 0, q_7)$. This structure is characterized by the occurrence of satellite peaks of *hkl* reflections for which (h + k = 2n + 1) [15]. The behaviour of these magnetic satellite reflections confirms that the transition at $T_{c/c} = 192$ K revealed in the magnetization curve of figure 2(a), corresponds to a magnetic phase transition from Fmc for $T > T_{c/c} = 192$ K to Fmi for $T < T_{c/c} = 192$ K. The component q_z of the wavevector dictates the phase difference between Mn spin along screw axis **c** equal to $2\pi(0, 0, q_z)(0, 0, 1) = 2\pi q_z$. The temperature dependence of the propagation vectors q_z has been derived from the Rietveld refinements and is presented in figure 13(b)(e.g. $q_z = 0.21$ at 3 K corresponding to a phase change angle of 76° and a cone semiangle of 52.2° with respect to the *c*-axis). In addition, the further increase in the intensity of the (112) reflection at 3 K compared with the (112) intensity at 60 K (see figures 12(e) and 13(a)) can be ascribed to the additional magnetic scattering resulting from magnetic ordering of the Pr sublattice below $T_{\rm C}^{\rm Pr} = 35$ K. The Mn sublattice moments in $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ continue to exhibit the Fmi state below $T_{\rm C}^{\rm Pr} = 35$ K, which together with the parallel coupling of the Pr sublattice moments with the Mn moments along the *c*-axis, results in the region of combined structures (Fmi + F(Pr)) as indicated in figure 2(a) and depicted in (figure 1(d)).

 $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$. Comparison of the diffraction patterns for $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ at 300, 100 and 7 K in figure 14, show several significant features. In particular, similar to the discussion for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$, the behaviour of the

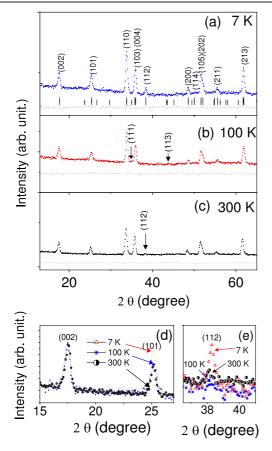


Figure 14. (a), (b), (c) Plots of Rietveld refinements to the neutron diffraction patterns of $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ at 7 K, 100 K and 300 K, respectively ($\lambda = 1.6660$ Å; MRPD, HIFAR). The lines show the calculated profiles through the experimental data and the residuals. The markers for the crystallographic (top) and magnetic (bottom) structures are indicated for the 7 K pattern in (a). Figures (d) and (e) show 2θ regions of interest at 7, 100 and 300 K on expanded scales for several peaks with magnetic scattering.

(112) and (101) reflections are consistent with ferromagnetic ordering at 300 K of interlayer spin components leading to the Fmc-type magnetic state. We conclude that Pr_{0.6}Lu_{0.4}Mn₂Ge₂ exhibits the Fmc structure between $T_{C}^{\text{inter}} = 321$ K and $T_{\rm N}^{\rm inter}$ = 172 K as indicated in figure 4(a). On the other hand the disappearance of the (112) reflection combined with the appearance of the (111) and (113) reflections at 100 K (figure 14(b)) indicates that the ferromagnetic ordering at 300 K of interlayer spin components has given way to antiferromagnetic coupling at 100 K, resulting in the AFmc structure depicted in figure 1(e) [7, 15]. Based on our magnetization and neutron diffraction results we therefore conclude that $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ exhibits the AFmc structure between $T_{\rm N}^{\rm inter}$ = 172 K and $T_{\rm C}^{\rm Pr}$ = 31 K as indicated in figure 4(a). The absence of magnetic scattering from the (111) and (113) reflections at 7 K, combined with the increase of the (112) and (101) peak intensities (figure 14(a)), indicates that the interlayer spin components of the Mn moments align parallel, again leading to an Fmc structure for the Mn sublattice. In noting the increased magnetization below $T_{\rm C}^{\rm Pr} =$ 31 K as in figure 4(a), the additional contributions from Pr moments coupled parallel to the Mn moments along the c-axis lead to the combined (Fmc + F(Pr))-type magnetic structure (figure 1(f)) for $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ below $T_C^{Pr} = 31$ K. It should be noted that with the limited resolution of the MPRD data no evidence for satellite reflections, indicative of the Fmi structure, were observed in figure 14. The absence of the Fmi structure in $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ is consistent with the behaviour of the $PrMn_2Ge_2$ ($q_z = 0.28$ at 2 K) and $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ ($q_z = 0.21$ at 3 K) compounds which demonstrate that q_z decreases with Lu substitution.

While it was not possible to obtain neutron diffraction patterns above room temperature for both samples, as mentioned above, we can none the less conclude that Pr_{0.8}Lu_{0.2}Mn₂Ge₂ and Pr_{0.6}Lu_{0.4}Mn₂Ge₂ exhibit the antiferromagnetic AFI structure (figure 1(a)) between 330 and 397 K (figure 2(b)) and 321-375 K (figure 4(b)) respectively. This conclusion is based on the analogues between Pr_{0.8}Lu_{0.2}Mn₂Ge₂ and PrMn₂Ge₂ [15, 26] as well as between SmMn₂Ge₂ [11–13] and Pr_{0.6}Lu_{0.4}Mn₂Ge₂ which both exhibit similar magnetic behaviours and the presence of the AFI state above T_C^{Inter} . Overall, combined with the magnetization results outlined above, the neutron investigations have enabled the magnetic structures of Pr_{0.8}Lu_{0.2}Mn₂Ge₂ and Pr_{0.6}Lu_{0.4}Mn₂Ge₂ to be The magnetic regions and their magnetic determined. structures are summarized in figures 2 and 4.

4. Conclusions

The magnetic phase transitions in $Pr_{1-x}Lu_xMn_2Ge_2$ (x = 0.2, x = 0.4) has been investigated and their magnetic structures determined. $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ displays antiferromagnetism below Néel temperature $T_N^{intra} = 397$ K transforming to three successive ferromagnetic phases below $T_N^{inter} = 300$ K, with ferromagnetic order of the Pr sublattice also occurring below $T_C^{Pr} = 35$ K (figure 2). $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ also displays antiferromagnetism below $T_N^{intra} = 375$ K, with decrease in temperature leading to successive ferromagnetic, antiferromagnetic and ferromagnetic phases and the re-entrant ferromagnetic behaviour shown in figure 4. Ferromagnetic order of the Pr sublattice also occurs below $T_C^{Pr} = 31$ K.

Substitution of Lu for Pr decreases the spacing of the Mn–Mn and Mn–Pr atoms (table 1), leading to a decrease of Mn moment itself at 4.5 K in addition to the dilution effect of Pr moment. The Debye temperatures have been derived to be 320 K and 400 K for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ and $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$, respectively. Below the Pr ordering temperature, T_C^{Pr} , the transferred hyperfine fields at the ⁵⁷Fe nucleus in both $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ and $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ are found to be smaller at 4.5 K than above T_C^{Pr} (figures 9(a) and 10(a)), thus establishing that the transferred hyperfine field from Pr is opposite to that produced by Mn. The magnetic structures for $Pr_{0.8}Lu_{0.2}Mn_2Ge_2$ and $Pr_{0.6}Lu_{0.4}Mn_2Ge_2$ are summarized in figures 2 and 4 respectively with the relevant magnetic structures depicted in figure 1.

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